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Revisiting the conformation of xanthan and the effect of industrially relevant treatments


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Revisiting the Conformation of Xanthan and the Effect of Industrially Relevant Treatments

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Abstract: The structure and conformation of xanthan in aqueous solution following various processing treatments typically encountered in its application were investigated in this study. Treatments such as heating, autoclaving, high pressure homogenization and irradiation were subjected to the same sample. Parameters such as weight average molecular weight ($M_w$), polydispersity index, root mean square radius of gyration, intrinsic viscosity and Huggins constant were used to monitor the effect of these treatments. Additionally, we have quantified the mass recovery of samples examined by gel permeation chromatography and light scattering to properly account for all fractions present in xanthan solutions. Atomic force microscopy (AFM) images together with height measurements confirmed that xanthan conformation is double helical ordered renatured state (pre-heat treated by the manufacturer) in dilute solution conditions and random coil conformation in very dilute solution. The ordered (renatured) conformation is shown to have partially molten double helix, with more flexibility than the perfectly ordered native double helix. Heat treatment for 2 hours at 85°C reduces the $M_w$ of xanthan to half its initial value, and mass recovery measurements indicate that it completely overcomes its associative nature. Thermally treated xanthan solution in the dilute region leads to an order-disorder transition, as determined by contour length per unit mass. Similarly, irradiation of xanthan solution results in an order – disorder transition together with the production of single strand low molecular weight molecules. Autoclaving and high pressure homogenisation treatments cause degradation of xanthan. The results from treated xanthan solutions following high pressure homogenization and irradiation confirm that xanthan does not reassociate. A revised summary of xanthan conformation in solution together with schematic models following the various treatments are proposed.
1 Introduction

Xanthan is an extracellular hetero-polysaccharide produced by the bacterium xanthomonas campestris. It is an anionic polysaccharide with cellulose backbone (β-1-4)-D glucose substituted at C3 on alternate glucose residues with a trisaccharide side chain (Morris, 2006). The noncarbohydrate substituents include O-acetate on the inner mannose residue and pyruvate on the terminal mannose residue. Different strains or fermentation conditions give rise to differing degrees of acetylation and pyruvlation, which moderate its functionality (Williams & Phillips, 2004). It is one of the major microbial polysaccharides employed in many industrial processes and often subjected to various processing conditions. Using X-ray diffraction, xanthan was shown to adopt a helical conformation having a five-fold symmetry and a pitch of 4.7 nm in the solid state (Moorhouse, Walkinshaw & Arnott, 1976). However, in aqueous solution, xanthan undergoes an ‘order’ (helix) to ‘disorder’ (coil) transition depending on temperature, ionic strength and pH (Capron, Brigand & Muller, 1997; Milas, Reed & Printz, 1996), as well as on acetyl and pyruvate content (Morrison, Clark, Talashek & Yuan, 2004; Smith, Symes, Lawson & Morris, 1981). The ordered molecule of xanthan can undergo a conformational transition to a disordered flexible coil either by increasing the temperature above characteristic temperature (T_m) or by decreasing the salinity of solution (Callet, Milas & Rinaudo, 1989; Capron, Brigand & Muller, 1997; Capron, Yvon & Muller, 1996; Gunning, Cairns, Kirby, Round, Bixler & Morris, 1998; Hacche, Washington & Brant, 1987; Kawakami & Norisuye, 1991; Mehyar, Liu & Han, 2008; Milas, Reed & Printz, 1996; Milas & Tinland, 1990; Oviatt & Brant, 1994b; Terada et al., 2005). T_m largely depends on concentration of the polymer and external salt (Rinaudo, 2001). The order-disorder conformational change (denaturation) of xanthan molecule in 0.01M NaCl is reported at temperature near 50°C (Capron, Brigand & Muller, 1997). Within the ordered structure there exist two different kinds of conformation - native and renatured. Xanthan fermentation broth without any heat treatment is reported to be in native state. Renatured xanthan can be obtained by heating the native form for few minutes at about 90°C with immediate fast cooling (Callet, Milas & Rinaudo, 1987; Capron, Brigand & Muller, 1997; Milas, Reed & Printz, 1996). The renatured conformation is believed to be double helical with more flexibility than native conformation (Capron, Brigand & Muller, 1997; Oviatt & Brant, 1994b). The disordered structure (denatured) is unanimously believed to be random coil and can be obtained by raising the temperature above T_m (>50°C) while prolonged heating (> 2
hours) at high temperature results in degradation (Capron, Brigand & Muller, 1997; Nishinari, Watase, Rinaudo & Milas, 1996). The stiffness of the xanthan polymer chain has been characterised by its persistence length $L_p$, which is reported to be 5 nm, 45 nm and 125 nm for flexible random coil, single helix and rigid double helical conformations, respectively (Rinaudo, 2001). However, the strandiness of xanthan can be characterised by contour length $M_L$, which is close to 1000 Dalton per nanometre (Da nm$^{-1}$) and 2000 (Da nm$^{-1}$) for single and double helix, respectively (Sato, Kojima, Norisuye & Fujita, 1984; Sato, Norisuye & Fujita, 1984).

Several treatments such as heating, autoclaving, microfluidising, etc. have been employed on polysaccharides including xanthan to overcome their associative nature and to promote conformational changes. Additionally, knowledge of the effect of these processes is beneficial for optimum performance of xanthan in various formulations. Capron, et al (1997) reported that the effect of heat treatment (110°C for 20 minutes in 0.1M NaCl) on xanthan conformation showed dissociation of the double helix into single strand i.e. denaturation (1997). Lagueynte and Paquin (1998) showed that the process of microfluidisation (high pressure homogenisation) on xanthan solution caused degradation due to mechanical stress. Evidence for degradation was deduced from the continuous fall in the molecular weight and reduction in shear viscosity. Similar degradation effects following high pressure homogenisation of starch / amylopectin, methylcellulose and chitosan were reported by Modig, et al. (2006), Fluory et al. (Floury, Desrumaux, Axelos & Legrand, 2002) and Kasai, et al., (2003) respectively. Autoclaving is another method used successfully to disperse aggregated molecules and thus reduce the associative nature of polysaccharides such as xyloglucan, oat β-glucan, and dextran in diluted solution (Wang, Wood, Cui & Ross-Murphy, 2001). Following autoclaving, the degree of aggregation was found to be moderately reduced for xyloglucan and dextran whereas for oat β-glucan aggregation was significantly reduced based upon evidence from intrinsic viscosity and molecular weight distribution profiles.

The effect of high energy radiation, such as gamma ray and electron beam, on dilute polysaccharide solutions has been widely reported to result in degradation (Fei, wach, Mitomo, Yoshii & Kome, 2000). Free radicals, such as hydroxyl radical, hydrated electron and hydrogen atom, are generated from the radiolysis of water which randomly attack the polymer chain causing scission (Al-Assaf, Phillips, Deeble, Parsons, Starnes & Von Sonntag, 1995; Gulez, Al-Assaf & Phillips, 2011).
The precise structural details of the ordered conformation of xanthan remains a subject of debate, as two different models have been proposed. One group of researchers suggested a single stranded structure for ordered xanthan stabilised by alignment and packing of side chains along the polymer backbone. This hypothesis was based on light scattering, viscometry, sedimentation and concentration independence studies (Callet, Milas & Rinaudo, 1987; Milas, Reed & Printz, 1996; Milas & Rinaudo, 1986; Milas & Tinland, 1990). On the other hand, the majority of studies have suggested a coaxial double helix for the ordered xanthan conformation (Brant, Hacche & Washington, 1985; Capron, Brigand & Muller, 1997; Gunning, Kirby & Morris, 1996; Hacche, Washington & Brant, 1987; Hatakenaka, Liu & Norisuye, 1987; Kawakami & Norisuye, 1991; Meyar, Liu & Han, 2008; Morrison, Clark, Talashek & Yuan, 2004; Oviatt & Brant, 1994a, b; Paradossi & Brant, 1982; Terada et al., 2005).

In this study, a commercial xanthan sample was subjected to different treatments in aqueous solution. These treatments were heating, autoclaving, high pressure homogenisation and radiation. Changes were monitored using light scattering and capillary viscometry. Crucially, in this study we quantified mass recovery on GPC-MALLS to properly characterise this heterogeneous system. Measurements based upon soluble fractions only cannot provide a full understanding of xanthan solution properties. Our objectives were: (i) To determine the corresponding effect on structural conformation in order to clarify some of the conflicting reports, (ii) to provide better understanding of how xanthan conformation could affect its functionality in various formulations and (iii) to examine whether xanthan order-disorder conformation is reversible following irradiation and autoclaving.

2 Materials and Methods

A commercial xanthan sample (food grade, Lot No 1394/FFA) in the powder form was kindly provided by Jungbunzlauer, Austria. Its pyruvate and acetyl contents were 6.3% and 2.2% respectively. The sample has already been processed by the supplier to remove cell debris. Distilled water was used for the dispersion of xanthan and all reagents were of analytical grade obtained from Fisher Scientific, UK.

2.1 Loss on drying and solubility

The loss on drying was determined according to a method described previously (Al-Assaf, Phillips, Aoki & Sasaki, 2007). The solubility was determined according to the following
method. The weight ($W_i$) of a 70 mm glass fibre paper (pore size 1.2 micron) is determined following drying in a convection oven (Sanyo, MOV-212F) at 105°C for 1 hour and subsequently cooled in a desiccator containing silica gel. A dispersion was prepared by adding xanthan powder to distilled water to a concentration of 0.5 wt% ($S$) while stirring, followed by overnight hydration at room temperature. The hydrated dispersion was then centrifuged for 2-5 minutes at 2500 rpm prior to filtration. Drying of the filter paper was carried out in an oven at 105°C followed by cooling to a constant weight ($W_f$). The percentage of insoluble material was calculated using:

$$\%\text{Insoluble} = \left(\frac{W_f - W_i}{S}\right) \times 100$$  \hspace{1cm} \text{[Equation 1]}

2.2 Solution preparation

An appropriate weight of xanthan was dissolved in distilled water and dispersed using a propeller type mixer at 200 rpm for 30 second. The solution was then left to hydrate overnight on a roller mixer. Appropriate dilution was then made in distilled water for the various treatments and measurements described below.

2.3 Heat treatment

Xanthan solutions at 4 mg ml$^{-1}$ in distilled water were introduced into sealed ampoules with Teflon caps and immersed in water bath thermostated at 85°C for different time periods (1 - 4 hours). The solutions after the various treatments were stored at 4°C until further use. Appropriate dilution was carried out for GPC-MALLS measurements.

2.4 Autoclaving

Xanthan solutions (4 mg ml$^{-1}$) were placed into sealed autoclave glass vials with metallic caps and placed into an autoclave chamber. Autoclaving treatment was performed at 120°C for 1 to 4 hours (Model BSB040, Astell Scientific, UK). Depending on the duration, the whole treatment cycle (heating-stabilising-cooling) took nearly 2-5 hours. The treated samples were cooled to room temperature, stored at 4°C and used within 24 hours of preparation. Appropriate dilution and sample pre-treatment (described in the relevant sections) were carried out for GPC-MALLS measurements.
2.5 High pressure homogenisation

Xanthan solutions at 4 mg ml\(^{-1}\) in distilled water were homogenised using a laboratory scale high pressure homogeniser (Nanomiser NM2-L100-D07, Collision type S generator, Yoshida Kikai Co. Ltd). The solution was passed 3 times through the Nanomiser at each pressure condition and stored as described above for GPC-MALLS measurements.

2.6 Irradiation

A \(^{137}\)Cs- gamma source supplied by Mainnance, UK (dose rate 36Gy/h) was used to irradiate a xanthan solution at 4 mg ml\(^{-1}\) in distilled water: 10 ml were transferred to pyrex tubes, saturated with N\(_2\)O for 10 minutes and then sealed with glass stoppers. The solutions were then irradiated at dose levels of: 0.9, 1.8, 2.7, and 3.6 kGy. Gray (Gy) is defined as the absorption of one joule of ionizing radiation by one kilogram of matter. Under these conditions 90% of the radicals generated from radiolysis of the water are the highly oxidising hydroxyl radicals (Al-Assaf, Phillips, Deeble, Parsons, Starnes & Von Sonntag, 1995). Appropriate dilution was carried out for GPC-MALLS.

2.7 GPC-MALLS system

The GPC-MALLS system consisted of a degasser ERC-3215a (ERC, Japan), a constametric \(\oplus\) 3200 MS pump (Thermo separation Products, FL), an injection valve with 100\(\mu\)l loop (Reodyne 7725i) fitted inside a temperature regulated oven (Gilson, Model 831, UK) and a DAWN-DSP multi-angle light scattering photometer (Wyatt Technology, Santa Barbara, CA, USA) equipped with He-Ne laser (\(\lambda = 633\)nm). Simultaneous concentration detection was performed using a calibrated differential refractometer (RI 2000, Schambek, Germany). A refractive index increment \(dn/dc\) value of 0.144 was used in the calculations (Capron, Brigand & Muller, 1997). The mass recovery was obtained from the ratio of mass eluted from column (determined by integration of the refractometer signal) to the mass injected. The mobile phase was 0.1M LiNO\(_3\) containing 0.005% NaN\(_3\) filtered through 0.2\(\mu\)m pore size cellulose nitrate membrane. The samples injected were subjected to prior filtration through a nylon filter of 0.45\(\mu\)m pore size. A set of two columns SB-803HQ & SB-806HQ (8mmx300mm, Shodex OHpak, Japan, exclusion limits 1x10\(^5\) and 2x10\(^7\) g/mol) was used for the separation. The flow rate for the eluent was 0.45 ml/min. The Berry fitting method with linear fit was used for data processing in ASTRA software (Version 4.90.08). For the measurement of molecular weight, the solutions after the various treatments (autoclaving,
high pressure homogenisation, and radiation) were further diluted into the range of 0.25 - 1 mg ml⁻¹ and allowed to hydrate on a roller mixer for an additional 5 hours. The solutions all contained the same molarity of buffer salt as the mobile phase. In the final preparation step the solutions were heated for 20 minutes at 60°C and injected while hot. The entire GPC-MALLS system was maintained at 60±1°C. Temperature control was achieved using the in-house heating methodology provided thermostatic heating to all pipework between the detectors.

The well known expression for radius of gyration of a rigid rod polymer (Sato, Norisuye & Fujita, 1984) used in previous studies to determine the strandness of xanthan is given in following equation:

\[
\left< S^2 \right>^{1/2} = \frac{1}{\sqrt{12}} \left( \frac{M_w}{M_L} \right) 
\]  

[Equation 2]

Where, \( \left< S^2 \right>^{1/2} \) is radius of gyration (\( R_g \)), \( M_w \) is molecular weight of polymer, \( M_L \) is molar mass per unit contour length of the rod. This can be obtained by plotting \( \frac{M_w}{R_g \sqrt{12}} \) on y-axis against \( M_w \) on x-axis and extrapolating to \( M_w = 0 \). At molecular weight approaching zero the polymer approximates a perfect rigid rod shape, while the intercept on y-axis will give the linear mass density for the polymer (Sato, Norisuye & Fujita, 1984).

2.8 Intrinsic viscosity

A calibrated Ubbelohde viscometer (Cannon Ubbelohde Semi-Micro Calib 75) was used to measure the flow time of solvent and solutions. Xanthan solution (at 4 mg ml⁻¹) was prepared in 0.1M LiNO₃ solution containing 0.005% NaN₃. The solutions after various treatments were stored at 4°C until further use. The viscometer was immersed in a water bath maintained at 60±0.1°C. The concentration of test solution was chosen such that for all these dilutions, the ratio of efflux time of solution to the solvent lies within the relative viscosity range of 2.2 to 1.2. All dilutions were made in situ. The efflux time at each dilution was measured in triplicate and the average was taken.

The intrinsic viscosity can be obtained by a double extrapolation to zero concentration using the combination of Huggins and Kraemer equations given below.

\[
\eta_{sp}/c = [\eta] + k_2[\eta]^2 c 
\]  

[Equation 3]
\(
\frac{\ln \eta_r}{c} = [\eta] - k_2 [\eta]^2 c
\)

[Equation 4]

Where \(\eta_r\) is the relative viscosity defined as the ratio of solution viscosity to solvent viscosity, and \(\eta_{sp}\) is the specific viscosity satisfying \(\eta_{sp} = \eta_r - 1\). \(k_1\) and \(k_2\) are Huggins and Kraemer constants, respectively. The Kraemer constant \(k_2\) is generally negative and smaller in magnitude than the Huggins constant \(k_1\) (Huggins, 1942; Kraemer, 1938; Yung & Shoemaker, 1991) and both are used to determine the quality of solvent.

The relationship of molecular weight to intrinsic viscosity and radius of gyration of a polymer is given by the Mark-Houwink-Kuhn-Sakurada formulae (Equations 5 & 6). The coefficients 'a' and 'c' determine the conformation of the polymeric chain (Tombs & Hardings, 1998). For compact spherical conformation the value of 'c' is ~0.33 and for random coil it is in the range of 0.5~0.6, while at a 'c' value~1.0, the conformation is rigid rod. Similarly, the value of 'a' for compact conformation is 0, and 0.5~0.8 for random coil and is 1.8 for a rigid rod conformation.

\([\eta]=K'M^a\)

[Equation 5]

\(R_g=K''M^c\)

[Equation 6]

### 2.9 Atomic Force Microscopy

Solutions of xanthan were prepared in distilled water according to the method described above. Additionally the solutions were heated at 60°C for 20 minutes to ensure the complete dissolution and diluted to a final concentration of 10 μg ml\(^{-1}\). Sample deposition took two forms. In the first, drops of the solution (5 μL) were applied to freshly cleaved mica surfaces, allowed to dry for about 10 min, and then imaged in air at room temperature. In the second, a modified mica surface was used. This was prepared by incubating freshly cleaved mica in a solution of poly-l-lysine (Sigma P8920) at 10 μg ml\(^{-1}\) for 1 minute followed by rinsing in ultrapure water and drying in a stream of argon. A 100 μL drop of the 10 μg ml\(^{-1}\) xanthan solution was incubated on a poly-l-lysine coated mica substrate for 1 minute at room temperature, after which the surface was rinsed by dispensing ultrapure water (5 x 1 ml) from a pipette. The sample was dried by blowing excess water off the mica with argon.

The atomic force microscope used in this study was an MFP-3D (Asylum Research, CA, USA) and was operated in the A/C control mode. The cantilevers used were Olympus AC160TS (Olympus, Japan) with a spring constant of around 42 Nm\(^{-1}\) oscillated at a frequency 10% below resonance (typically around 320 kHz). The same probe was used to collect all of the images presented in this study. The damping set point for imaging was kept...
to the minimum value that allowed stable tracking of the sample surface to minimise any sample deformation. Images were acquired at a scan rate of 1 Hz.

3 Results and discussion

3.1 Conformation of xanthan

Figure 1 shows a typical AFM image obtained on the sample prepared by drop deposition. The relatively linear appearance of the molecular chains seen in the image suggests that they are rigid. Height measurements indicate that the polymer chain height in thicker regions is 1.883 nm which is quite close to the helical diameter of xanthan (~2 nm) reported in literature (Gunning, Kirby & Morris, 1996), while it is almost half (0.886nm) in thinner regions suggesting the presence of molten double helical structure.

Figure 1. Topography image of xanthan solution by AFM using A/C tapping mode and drop deposition method; height measurement of xanthan helices to determine strandedness in (A) thicker region and (B) thinner region.

Figure 2 shows a typical AFM image of the sample obtained following deposition of the dilute xanthan solution onto poly-l-lysine coated mica. The molecular chain is strikingly less pronounced in terms of height (0.178 nm) than those seen in figure 1 and, in addition, the sharp kinks along the chain suggest that it is significantly more flexible.
Figure 2. AFM topography image of xanthan deposited onto poly-l-lysine coated mica; height measurement to determine molecular conformation.

The deposition methodologies differ in two important respects: Incubation of dilute xanthan solution onto poly-l-lysine coated mica will result in almost immediate electrostatic binding of the negatively charged xanthan molecules onto the positively charged poly-l-lysine layer, thus trapping the solution conformation of some of the molecules on the mica, and any remaining unbound molecules will be removed by the water rinse. Drop deposition, on the other hand, causes concentration of the xanthan solution as the solvent evaporates. Furthermore, the molecules are likely to remain mobile and hydrated until most of the water is lost, at which point the molecules finally become trapped on the mica surface by physisorption. The difference in the AFM images obtained using the different methods quite clearly point to conformational differences of the xanthan in the resulting molecular deposits. In the case of the drop deposited sample (Figure 1) we propose that the molecular mobility coupled with the increasing concentration of the xanthan induces the renatured helical conformation once dried. The electrostatically bound xanthan molecules, such as the example presented in Figure 2, appear to be trapped in the random coil conformation. This is consistent with the fact that they were bound to the surface from a very dilute solution of negligible ionic strength, where the coil conformation would be expected to predominate. Thus, the AFM images presented in Figures 1 and 2 provide confirmation of the existence of two of xanthan’s possible solution conformations.

3.2 Effects of heat treatment

The measurements to determine the weight average molecular weight ($M_w$), radius of gyration ($R_g$), mass recovery and polydispersity ($M_w/M_n$) were carried out at 60°C to ensure 11
aggregate free solutions. Similar measurements were previously reported at 25°C by other researchers (Capron, Brigand & Muller, 1997; Nishinari, Watase, Rinaudo & Milas, 1996). Xanthan was shown not to lose its double strand suddenly at high temperature, but rather partial melting of the double helix. Furthermore, previous studies also suggest that even after heating xanthan solution at 95°C for 2.5 hrs, the dimer (double helical) fraction present is more than 60% (Kawakami & Norisuye, 1991; Mehyar, Liu & Han, 2008). Based on these evidences, we do not expect to lose the strandedness of xanthan solution completely by measuring molecular weight at 60°C (as the elution time is ~1 hour in the GPC columns); nevertheless this temperature might have an effect on the flexibility of xanthan molecule in solution. Figure 3 shows a typical elution profile monitored by the refractive index detector together with the molecular weight of each fraction plotted as a function of elution volume. No irregularity can be observed in the elution profile suggesting that the conditions are adequate for the smooth elution of high molecular weight fractions followed by low molecular weight.

The molecular weight for the solution heated for 20 min is given in Table 1 and the value obtained is consistent with previously reported values (Capron, Brigand & Muller, 1997; Milas, Reed & Printz, 1996).

<table>
<thead>
<tr>
<th>Time (min)</th>
<th>$M^*$ $10^4$ (Da)</th>
<th>$R_g$ (nm)</th>
<th>$M_n/M_n$</th>
<th>%Mass Recovery</th>
<th>$[\eta]$ (ml g$^{-1}$)</th>
<th>$k'$</th>
</tr>
</thead>
<tbody>
<tr>
<td>20</td>
<td>21.7</td>
<td>140</td>
<td>1.64</td>
<td>83</td>
<td>2.065</td>
<td>0.62</td>
</tr>
<tr>
<td>30</td>
<td>20.9</td>
<td>144</td>
<td>1.54</td>
<td>80</td>
<td>2.112</td>
<td>0.49</td>
</tr>
<tr>
<td>40</td>
<td>24.9</td>
<td>153</td>
<td>1.50</td>
<td>80</td>
<td>2.066</td>
<td>0.65</td>
</tr>
<tr>
<td>60</td>
<td>25.8</td>
<td>165</td>
<td>1.29</td>
<td>96</td>
<td>2.115</td>
<td>0.43</td>
</tr>
<tr>
<td>120</td>
<td>14.8</td>
<td>144</td>
<td>2.75</td>
<td>100</td>
<td>2.093</td>
<td>0.60</td>
</tr>
</tbody>
</table>

Table 1 GPC-MALLS parameters and intrinsic viscosity data for thermally treated xanthan solutions.
Figure 3. Molecular weight versus elution volume and refractive index elution profile of xanthan. Injected at 0.25 mg mL$^{-1}$ at the conditions described in section 2.7.

The mass recovery was 83% suggesting that ~ 17% of the mass was in the aggregated form and was retained on a 0.45 μm filter. Independent measurement to determine the solubility identified the presence of insoluble materials (aggregates) of nearly 8% in xanthan solution which was retained on a 1.2 micron filter. This suggests the disassociation (opening up) of aggregates present in xanthan solution following filtration through smaller pore size filters, and is consistent with results reported for hyaluronan (Al-Assaf, Phillips, Gunning & Morris, 2002) and gum Arabic (Al-Assaf, Sakata, McKenna, Aoki & Phillips, 2009). The intrinsic viscosity value (2.06 ml mg$^{-1}$) is similar to a previously reported value (Young & Shoemaker, 1991) and the Huggins constant value confirms good solvent condition (Higiro, Herald, Alavi & Bean, 2007; Morris, 1995; Morris, Cutler, Ross-Murphy & Rees, 1981). The effect of heat treatment (heating at 85°C) was investigated as described above. Immediately after the heating cycle was completed, the hot solution was injected into GPC-MALLS system. Heating for 30, 40 and 60 min gave comparable results with the exception of changes in the polydispersity and mass recovery. It is interesting to note that while there was no change in the molecular weight the mass recovery increased to 96% upon heating for 60 min and a lower polydispersity value was obtained. The reduction in the polydispersity value suggests disassociation of the high molecular weight aggregates to the extent where they can pass through a 0.45 micron filter. Further heating for 120 minutes shows a molecular weight reduction to nearly half the initial value of the 20 minute heated sample, but with 100% mass
recovery and the $R_g$ value remains more or less unchanged. Capron et al (1997) also reported the heat treatment of xanthan (30 minutes at 110°C). Their results also showed a reduction of molecular weight to half the initial value without affecting $R_g$, and they attributed this change to dissociation of double helix into single strand. However, no information was provided on the polydispersity and mass recovery. In this study we provide further evidence of dissociation following heat treatment at 85°C supported by the determination of the mass per unit length ($M_L$) value as shown in Figure 4. A value of $M_L$ for thermally treated xanthan was determined to be 904 Dalton/nm which is close to the value for single strand (1000 Dalton/nm) (Capron, Brigand & Muller, 1997; Terada et al., 2005). The overall effect of heat treatment on xanthan solution could be summarised as dissociation of double helix into single strands as shown in Figure 5.

![Figure 4. Contour length ($M_L$) plotted as function of molecular weight for xanthan following (*) heating; (△) high pressure homogenisation; (□) autoclaving; and (○) irradiation.](image)

Heat treatment
- Heating: $y = 151.25x + 898.62$
- $R^2 = 0.8887$
- HPH: $y = 119.81x + 1787.1$
- $R^2 = 0.982$
- Autoclave: $y = 142.30x + 1366.7$
- $R^2 = 0.9971$
- Radiation: $y = 155.88x + 1176.8$
- $R^2 = 0.995$

Heat Treatment

Renatured (expanded double helix with flexibility)

Chain dissociation
- $M_w$ reduce to half of original value
- but $R_g$ and $[\eta]$ remain unchanged and $M_L \approx 100$ Da/nm
- (opening up of double helix to single strand)
Figure 5. Schematic illustration of structural and conformational changes in xanthan aqueous solution following heat treatment.

Mark-Houwink Coefficients provide useful information concerning the conformational characteristics of polymer in solution. The slope of the curve obtained was ~0.7 (data not shown) which is less than the value (~1) for the perfectly rigid-rod double helical ordered native xanthan (Capron, Brigand & Muller, 1997; Terada et al., 2005). The result indicates that the test material is in a double helical ordered form, with greater flexibility compared to native xanthan in agreement with AFM images shown in Figure 1. Possible reasons for the slight variation from a perfect rigid-rod value could be due to the heat treatment and subsequent cooling or the underestimation of the intrinsic viscosity values determined by capillary viscometry. A previous study on cross-linked hyaluronan (hyalan A) reported the major effect of shear rate applicable in the capillary viscometer compared to zero shear intrinsic viscosity obtained from a low shear rate Contraves rheometer (Al-Assaf, Phillips, Gunning & Morris, 2002). A value of 8.18 ml mg⁻¹ was obtained from zero shear rate viscosity measurements compared to a value of 5.14 ml mg⁻¹ using a capillary viscometer.

3.3 Effects of high pressure homogenisation

Xanthan solution at 4 mg ml⁻¹ was subjected to high pressure homogenisation treatments at 25, 50, 75 and 100MPa in order to reduce its associative nature. The solution (4 mg ml⁻¹) was passed through the homogeniser three times and was further diluted to 0.25 mg ml⁻¹ for GPC-MALLS measurements. Its intrinsic viscosity was also measured at 60°C to correlate with light-scattering data. The values obtained are presented in Table 2.

Table 2. GPC-MALLS and viscometry data obtained for xanthan solutions following high pressure homogenisation treatment (3 passes at 25 MPa to 100 MPa).

<table>
<thead>
<tr>
<th>Treatment (MPa)</th>
<th>pressure (Da) $M_w$*10⁻⁵</th>
<th>$R_g$ (nm)</th>
<th>$M_w/M_a$</th>
<th>%Mass Recovery</th>
<th>$[\eta]$ (ml mg⁻¹)</th>
<th>$k'$</th>
</tr>
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<tbody>
<tr>
<td>Control</td>
<td>21.7</td>
<td>140</td>
<td>1.64</td>
<td>83</td>
<td>2.06</td>
<td>0.73</td>
</tr>
<tr>
<td>25</td>
<td>14.31</td>
<td>124</td>
<td>1.57</td>
<td>86</td>
<td>1.39</td>
<td>0.63</td>
</tr>
<tr>
<td>50</td>
<td>8.80</td>
<td>90</td>
<td>1.25</td>
<td>86</td>
<td>0.94</td>
<td>0.41</td>
</tr>
<tr>
<td>75</td>
<td>7.18</td>
<td>77</td>
<td>1.42</td>
<td>85</td>
<td>0.78</td>
<td>0.34</td>
</tr>
</tbody>
</table>
Table 2 shows a continuous fall in $M_w$, $R_g$, polydispersity and $[\eta]$ which indicates degradation of xanthan by high pressure homogenisation. The results further support the findings of a previous study on xanthan which reported polymer degradation due to mechanical stress, using reduction in molecular weight as the only evidence (Lagoueyte & Paquin, 1998). The results given in Table 2 also show a clear trend in increasing the mass recovery as a function of the applied total pressure. However, we found no evidence for dissociation of xanthan polymer chains (double strand to single strand), as $M_L$ remains ~2000 Dalton/mm (Figure 4). The results given in Table 2 are summarised schematically in Figure 6 where the restructured xanthan is degraded into smaller molecules, but retains its double helical conformation.

![Diagram](image)

**Figure 6.** Schematic illustration of structural and conformational changes in xanthan aqueous solution following high pressure homogenisation.

Conformational changes in xanthan solutions subjected to elevated temperatures are reported to be permanent and no reassociation takes place, once the dissociation has occurred. This has been shown by Capron, et al. (1997) by comparing the molecular weight of a sample obtained by injecting immediately after heat treatment, with that of the same sample stored at 4°C and injected the following day. In this study similar measurements were performed to show that conformational changes due to high pressure treatments on xanthan solution are permanent and time-independent. In order to demonstrate this effect, a high pressure homogenised solution (50 MPa, 3 passes) was stored at 4°C for 3 days and re-injected into GPC-MALLS to investigate any conformational changes. The results showed a slight reduction in $M_w$ from $8.80 \times 10^6$ to $8.6 \times 10^6$ (g/mol) while $R_g$, polydispersity and mass recovery remained unchanged. This suggests that the dissociation induced by high pressure treatment is
irreversible and xanthan molecules, albeit in double strand form, do not reassociate upon storage at 4°C.

### 3.4 Effects of autoclaving

The molecular weight parameters, intrinsic viscosity and Huggins constant of control and autoclaved xanthan solutions are given in Table 3. A contour plot to determine the conformation of the material is shown in Figure 4.

The autoclaving effect shows a continuous reduction in molecular weight, $R_g$ as well as intrinsic viscosity (Table 3). It demonstrates that the effect of autoclaving on xanthan is similar to the degradation seen on oat β-glucan in the study by Wang et al (2001). However, the linear mass density calculated for autoclaved xanthan shows a value of ~1500 Dalton/nm (Figure 4), which lies between the values for single strand (1000) and double strands (2000). So, it can be suggested that dissociation (due to high temperature) and degradation (due to high pressure) both occur simultaneously in xanthan solutions which are autoclaved. The results given in Table 3 are summarised schematically in Figure 7 where both degradation and dissociation of the renatured xanthan chains occur.

**Table 3.** GPC-MALLS and viscometry data obtained for xanthan solution after autoclaving (120°C, 1-4 hrs).

<table>
<thead>
<tr>
<th>Autoclave Time (hr)</th>
<th>$M_r\times10^{-5}$ (Da)</th>
<th>$R_g$ (nm)</th>
<th>$M_w/M_n$</th>
<th>%Mass Recovery</th>
<th>$[\eta]$ (ml mg$^{-1}$)</th>
<th>$k^1$</th>
</tr>
</thead>
<tbody>
<tr>
<td>Control</td>
<td>21.7</td>
<td>140</td>
<td>1.64</td>
<td>83</td>
<td>2.06</td>
<td>0.73</td>
</tr>
<tr>
<td>1</td>
<td>12.5</td>
<td>117</td>
<td>1.31</td>
<td>89</td>
<td>1.61</td>
<td>0.54</td>
</tr>
<tr>
<td>2</td>
<td>10.3</td>
<td>105</td>
<td>1.40</td>
<td>92</td>
<td>1.36</td>
<td>0.49</td>
</tr>
<tr>
<td>3</td>
<td>8.8</td>
<td>95</td>
<td>1.16</td>
<td>95</td>
<td>1.27</td>
<td>0.35</td>
</tr>
<tr>
<td>4</td>
<td>8.2</td>
<td>94</td>
<td>1.28</td>
<td>96</td>
<td>0.90</td>
<td>0.51</td>
</tr>
</tbody>
</table>
3.5 Effects of irradiation in dilute solution

Xanthan was irradiated in diluted state (4 mg ml⁻¹, solution saturated in N₂O) and further characterized by light scattering and viscometry. The results showed a reduction in molecular weight, intrinsic viscosity as well as $R_g$ values (Table 4). This indicates that irradiation of dilute solution results in degradation due to the generation of hydroxyl radicals (resulting from water radiolysis). These radicals randomly attack the polymer chains and lead to chain scission (Fei, wach, Mitomo, Yoshii & Kome, 2000). The generation of radicals and the energy delivered by the radiation might also affect the association of xanthan molecules, depending on the radiation dose and may lead to order-disorder transition. It is interesting to note that there was an increase in the polydispersity as a function of increasing the irradiation dose, consistent with a random attack mechanism on xanthan molecules. Furthermore, there was also an increase in the mass recovery, and at the highest radiation dose the value obtained suggests the complete dispersal of aggregates.

Table 4. GPC-MALLS and viscometry data obtained for dilute xanthan solutions (4 mg ml⁻¹) following irradiation treatments.

<table>
<thead>
<tr>
<th>Radiation dose (kGy)</th>
<th>$M_w$ $\times 10^5$ (Da)</th>
<th>$R_g$ (nm)</th>
<th>$M_w/M_n$</th>
<th>%Mass Recovery</th>
<th>$[\eta]$ (ml mg⁻¹)</th>
<th>$k'$</th>
</tr>
</thead>
<tbody>
<tr>
<td>Control</td>
<td>21.7</td>
<td>140</td>
<td>1.64</td>
<td>83</td>
<td>2.06</td>
<td>0.73</td>
</tr>
<tr>
<td>0.9</td>
<td>10.26</td>
<td>95</td>
<td>2.03</td>
<td>80</td>
<td>1.78</td>
<td>0.55</td>
</tr>
<tr>
<td>1.8</td>
<td>7.87</td>
<td>97</td>
<td>2.29</td>
<td>86</td>
<td>1.48</td>
<td>0.53</td>
</tr>
<tr>
<td>2.7</td>
<td>7.16</td>
<td>90</td>
<td>2.33</td>
<td>96</td>
<td>1.47</td>
<td>0.64</td>
</tr>
</tbody>
</table>
Further evidence for a radiation-induced order-disorder transition was obtained by linear mass density calculation. The $M_L$ value for an irradiated dilute xanthan solution is ~1173 Dalton/nm (Figure 4) which is close to the value for single strand. These results indicate that irradiation of xanthan in dilute solution leads to an order-disorder transition (dissociation followed by degradation). The overall effect of radiation on dilute xanthan solution could be summarised as *in situ* dissociation followed by degradation (Figure 8).

![Image](image-url)

**Figure 8.** Schematic illustration of structural and conformational changes in xanthan following irradiation in dilute aqueous solution.

To demonstrate that radiation-induced conformational changes of xanthan solution are permanent and time-independent, an irradiated solution (900 Gy dose) was stored at 4°C for 7 days and re-injected into GPC-MALLS. The data obtained is shown in Table 5 which demonstrates that the dissociation induced by such treatments is irreversible and xanthan molecules do not reassociate with time upon storage at 4°C.

**Table 5.** Molecular weight parameters for irradiated xanthan in aqueous solution and following storage for one week at 4°C.

<table>
<thead>
<tr>
<th>Treatment</th>
<th>Dose</th>
<th>$M_w \times 10^{3}$ (Da)</th>
<th>$R_g$ (nm)</th>
<th>$M_w/M_n$</th>
<th>%Mass Recovery</th>
</tr>
</thead>
<tbody>
<tr>
<td>Radiation</td>
<td>900Gy</td>
<td>10.26</td>
<td>95</td>
<td>2.03</td>
<td>80</td>
</tr>
<tr>
<td>After one week</td>
<td>10.10</td>
<td>98</td>
<td>2.07</td>
<td>80</td>
<td></td>
</tr>
</tbody>
</table>
4 Conclusion

Different treatments such as heating and ultrasonification have been applied to xanthan solutions in the past to break the molecular associations, but the precise changes induced by such treatments remain obscure. This is the first comprehensive study to apply different treatments (homogenisation, autoclaving, heat treatment and irradiation) on the same sample. Optimum conditions for GPC-MALLS have been demonstrated, yielding physical data on the molecular state of xanthan in solution that should be more reliable. Based on AFM, $M_w$, $R_g$, [$\eta$], Mark-Houwink coefficient $c$ and $M_L$ data we have confirmed that the test material is double helical ordered renatured state. Together with the results obtained in this study and those reported previously it is now possible to reconcile the two conflicting reports about xanthan conformation as shown in the scheme presented in Figure 9. Subsequently, the resulting conformations are proposed following each treatment based on the model given in Figure 9.

![Diagram](image)

**Figure 9.** Schematic representation summarising the order-disorder conformation of xanthan in aqueous solution reported in the literature. References as given in the introduction.

This work provides a rational basis for the manipulation of xanthan to overcome the barrier of its associative nature and aggregation. Furthermore, the effect of these treatments on xanthan conformation will also be useful to understand its synergetic interaction with other...
polysaccharides, particularly galactomannans. Finally, the study reveals how a series of low molecular weight xanthan fractions with tuneable conformations can be produced through the knowledge-led application of the treatments described herein.
5 References


